



The Vital Role of Ultra Thin Silicon Oxide Gate Dielectrics

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Abstract: In dry thermal oxidation process with Si, the growth rate of the initial parts of the reaction is currently modeled with additional terms; besides the usual linear parabolic was time dependent. However, due to limitation in the process set up, it was found in literature and in patent; these methods did not produce oxides less than 1.5 nm thick, often assumed equal to the native thickness. Attempt made to grow oxides in a furnace with dry oxygen, including cycles of cleaning and annealing in Ar. The effects limit the thickness to about 1.5 nm was investigated. These oxides were evaluated for the resulting quality of the interface of the oxides, with low temperature layer-by-layer methods.

Key words: Nanotransistor • Ultra thin film • Gate dielectric • Silicon oxide

INTRODUCTION

Nanotechnology is already a vast field and nano structured materials have been a central area of interest for quite some time in the scientific and technical communities. Many physical, chemical and biological reactions have been discovered to the self-organizing, or self-limiting. Nowadays, this has become one of the main areas of interest for invention; bottom-up synthetic procedures for the fabrication of nano structured functional materials.

Moreover, the last two decades, semiconductors come to be used in a wide range of electronic devices, such as MOSFETs (Metal-Oxide-Semiconductor-Field-Effect-Transistors), switching devices, voltage regulation, photocells and photo detectors. The success of the device is attributable to several technology important factors [1 - 3]. First, silicon can be thermally oxidized to produce a stable oxide which is an excellent insulator. Second, the surface state density at the silicon-oxide interface is sufficiently low to ensure reproducibility. Third, being planner, the structure is amenable to large-scale integration.

The present studies were performed with AES (Auger Electron Spectroscopy) which is particularly useful with silicon surfaces and reactions thereon, because the surface electronic properties and their relation to the atomic structure of the surfaces may be monitored in a very detailed way with these methods. More detail information is available in literature [3].

Experimental Procedure and Results: Mirror-polished n-type Si (100) is used as substrates. This wafer is cut into pieces with dimension of $3 \times 1 \times 0.2$ cm and then silicon sample are rinsed with ethanol in an ultrasonic bath. The AES chamber background pressure was 2×10^{-9} Torr. Keep in mind that the pressure inside the vacuum chamber increased during the oxide growth on silicon substrate. The Si structures are kept constant at 500°C.

Indeed, electrons of energy 3-20 keV are incident upon a conducting sample. These electrons cause core electrons from atoms contained in the sample to be ejected resulting in a photoelectron and an atom with a core hole. The atom then relaxes via electrons with a lower binding energy dropping into the core hole. The energy thus released can be converted into a X-ray or emit an electron. This electron is called Auger electron. The energy of the Auger electron is characteristic of the element that emitted and can thus be used to identify the element. The short inelastic mean free path (IMFP) of Auger electrons in solids ensures the surface sensitivity of AES. Although this technique can determine the composition of the top few layers of a surface and cannot detect hydrogen or helium. It is sensitive to all other elements, being most sensitive to elements with low atomic number. Normally, sample may be coated with carbon and oxygen during experiments and should thus be removed (usually by sputtering) for growing a pure thin oxide on silicon substrate. Furthermore, sputtering involves directly a beam of ions (usually Ar ions) on samples in the range of 500 to 5000 eV. This process

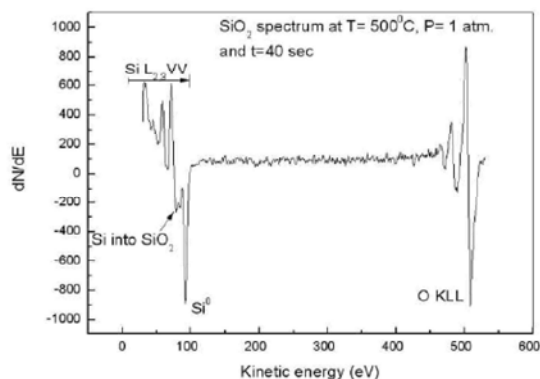


Fig. 1: Illustrative spectrum of Auger sputter depth profile of SiO_2 on $\text{Si}(100)$

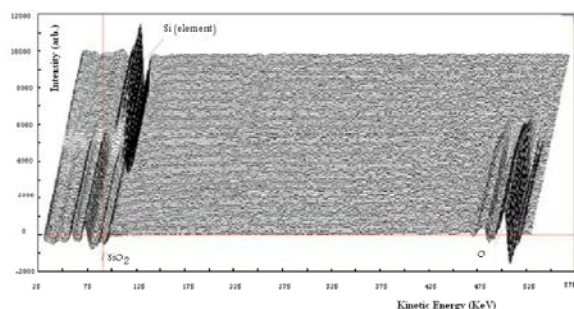


Fig. 2: Auger spectra from silicon oxide

cleans the surface, but also used to erode away the sample to reveal structure beneath the surface. In addition, the Auger electrons can escape the solid if the target atom is very near the surface. The escape depth of Auger electrons of Si (KLL) and O (KLL) are 5.5°A [1] and $5\text{-}15^\circ\text{A}$ [2] independently on energy of the incident electron beam, i.e. Auger electrons are only emitted from the outermost atomic layers. Hence, the energy of these electrons is counted and resulted an spectrum of number of electrons of the Auger peaks to identify the elements present in the sample, as it was found before [3-8] (Figures 1, 2).

Figure 1 shows an Auger sputter depth profile of SiO_2 on $\text{Si}(100)$. Auger measurements were alternated with sputtering and the sputter time varies along the direction of the plots. The sample was sputtered for 30 seconds between measurements. Additional information is available in published literature from the author's recent work [4-6]. In addition, Figure 2 illustrates the SiO_2 spectrum peaks at 62 and 78 eV; the Si, O peaks are due to reduction of SiO_2 by the electron beam.

The Auger process is quite remarkable by the tremendous range of energy over which this phenomenon is observed. Using the first order derivative of the signal

it is possible to get a better line assignment, because of the maximum peak at special energy where the elements present, generates a resonance like structure in $dN(E) / d(E)$, whose most negative excursion at this energy corresponds to the steepest slope of the $N(E)$.

DISCUSSION

The characteristic Auger spectra from silicon oxide and silicon are shown in Figures 2 and 3. The signal from each element has a characteristic shape and is determined using reference and moreover, Plasmon peaks can be distinguished from Auger peaks by their layer width as well. As shown in these figures, the typical sharp of minima of Si and O at 92 eV and 510 eV, respectively. These peaks followed by series of smaller peaks, the latter being due partly to other Auger components. The 92 eV peak is due to elemental silicon, but most of the following two peaks are due to silicon in silicon oxide. In fact, a steam oxidized sample produced only the last two peaks and no 92 eV peak. However, electron beam bombardment caused sufficient decomposition to produce some elemental silicon.

Figure 2 depicts the sputtering spectrum silicon with using 5 keV energy of Ar ion (Ar^+). It would help one to get further information about the specimen. Indeed, one is able to drill a hole in the specimen at some specified spot and clean the surface as well. For this purpose one can plot the signal strengths of selected Auger transition as a function of the sputtering time (Figures 3, 4 and 5). It is so important to choose the size of the Ar^+ beam spot for rastering a crater with a flat boom surface. The small-spot focused-beam provides a good solution for a wide range of applications. As one can see in these figures (Figures 1 and 2), the Si into SiO_2 , Auger signal decreases considerably at near interface and then the Plasmon peak at 75 eV increase to a high value corresponds to the Si substrate. More detail information reported in the literature [9-15].

Although departure from linear-parabolic kinetics [16] in the functional dependence, however, is the same over a large parameter range. An initial linear dependence with respect to time; follows by an extensive parabolic regime. It means that evolution of concentration is time-dependence and so the gradient of oxidant species is not constant. The enhanced initial oxidation rate (Figure 3) has been described earlier [17-19] as an exponentially decaying extra rate them, together with detailed extraction of relevant activation energies.

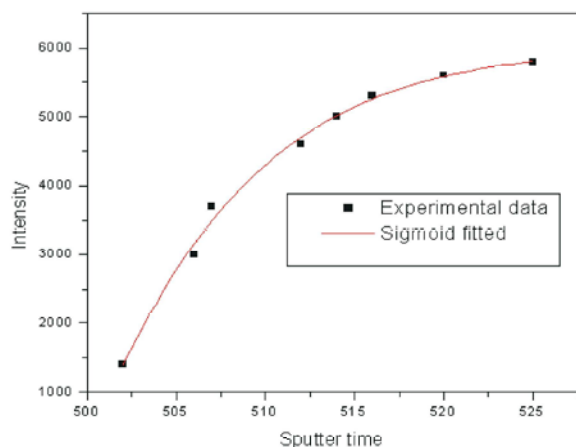


Fig. 3: The oxygen Intensity versus the sputter time.

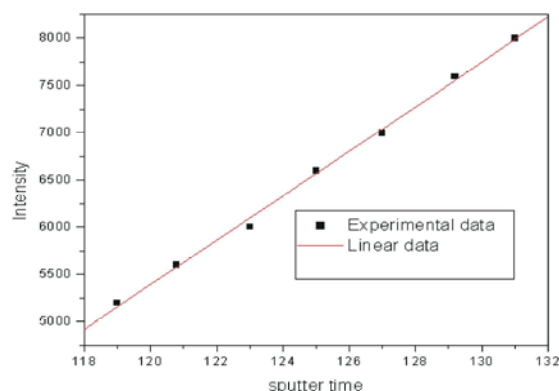


Fig. 4: The silicon intensity versus the sputter time

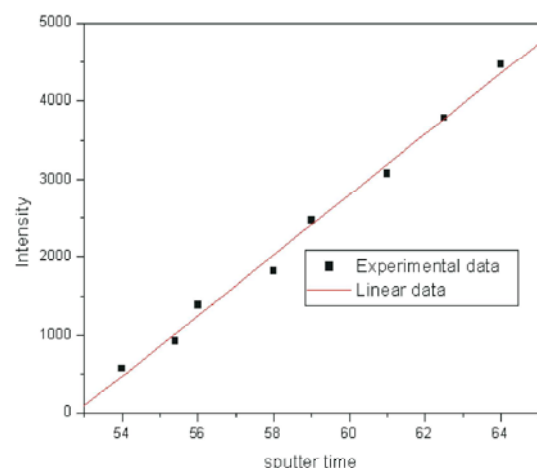


Fig. 5: The silicon oxide intensity versus the sputter time.

CONCLUSION

The vital role of very thin dielectrics is well defined in nano-electric field. The necessary attempt and motivation for characterization of silicon oxide in the thin regimes preformed. In the dry thermal oxidation process with Si the

growth rate of the initial parts of the reaction is currently modeled with additional terms, besides the usual linear-parabolic time dependency. It was great achievement to find the similar oxide growth as Deal-Grove / Massoud predicted, but it is as limiting case emerging from the Boltzmann-like expression.

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